

High temperature corrosion problems during gasification of biomass and waste

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Background - gasification



- Ideally: Use any carbon source for producing synthesis gas (CO/H_2)
- Use as raw material or use for heat/power production

Field exposures, Värö

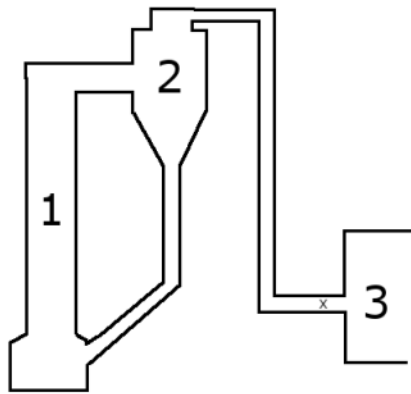


- Forest based feedstock
- CFB process
- Traditional bed material

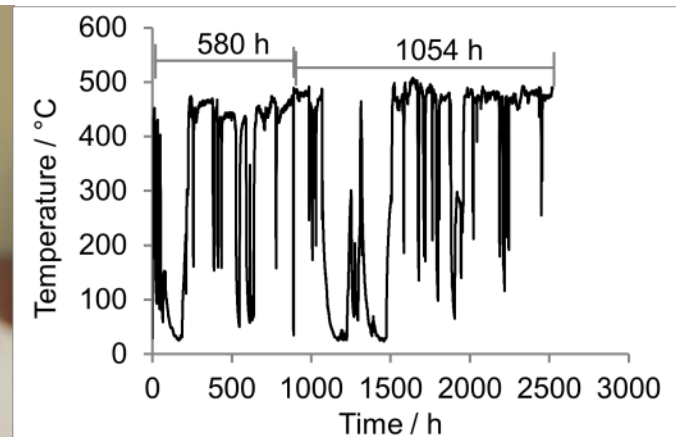
	H ₂	CO ₂	H ₂ O	CH ₄	CO	Remarks
Värö	10-15	10-15	10-15	5	15-20	Air blown

Exposures from Värö - experimental

- Three exposures were performed on different occasions in a non-cooled probe at the inlet of the lime kiln
 - One exposure 20 min (4 min above 350°C)
 - One intermediate exposure (580 h above 350°C)
 - One long-term exposure (1054 h above 350°C)



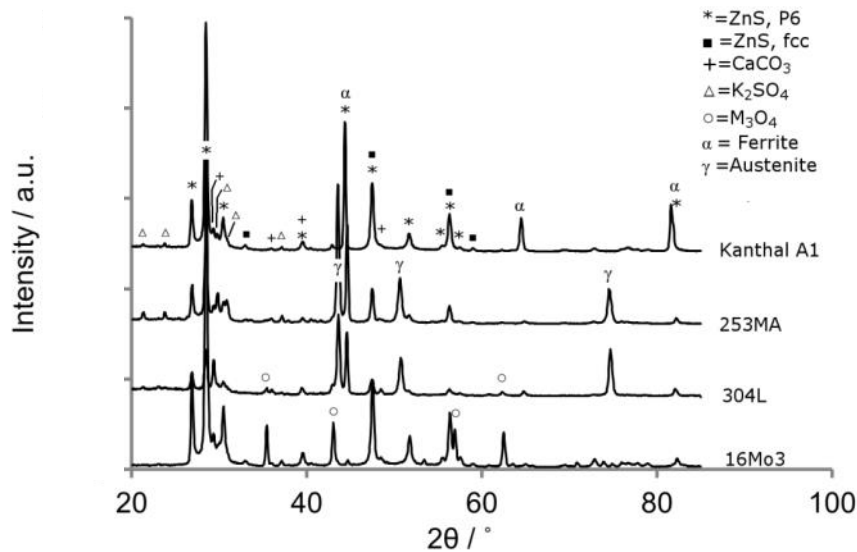
1: Gasifier, 2: Cyclone, 3: Lime kiln
X: Position of corrosion probe



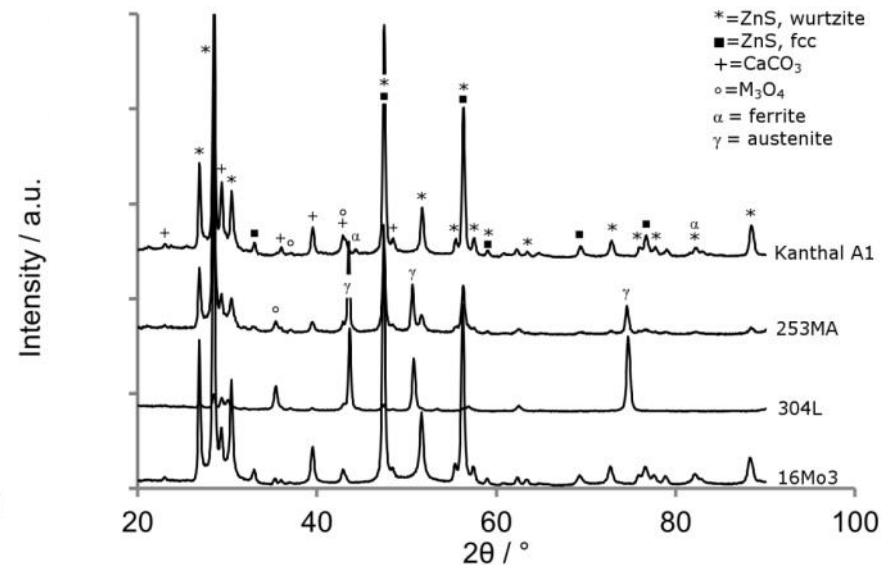
Exposed materials

Material	Fe	Cr	Ni	Si	Mn	Övrigt
16Mo3	Bal	-	-	-	0.4-0.9	Mo 0.25-0.35
253MA	Bal	20.75	11.11	1.61	1.44	Ce
304L	Bal	18.25	8.05	0.28	1.63	
Kanthal A1	Bal	21.2	-	0.18	0.08	Al 5.6

Results: XRD results



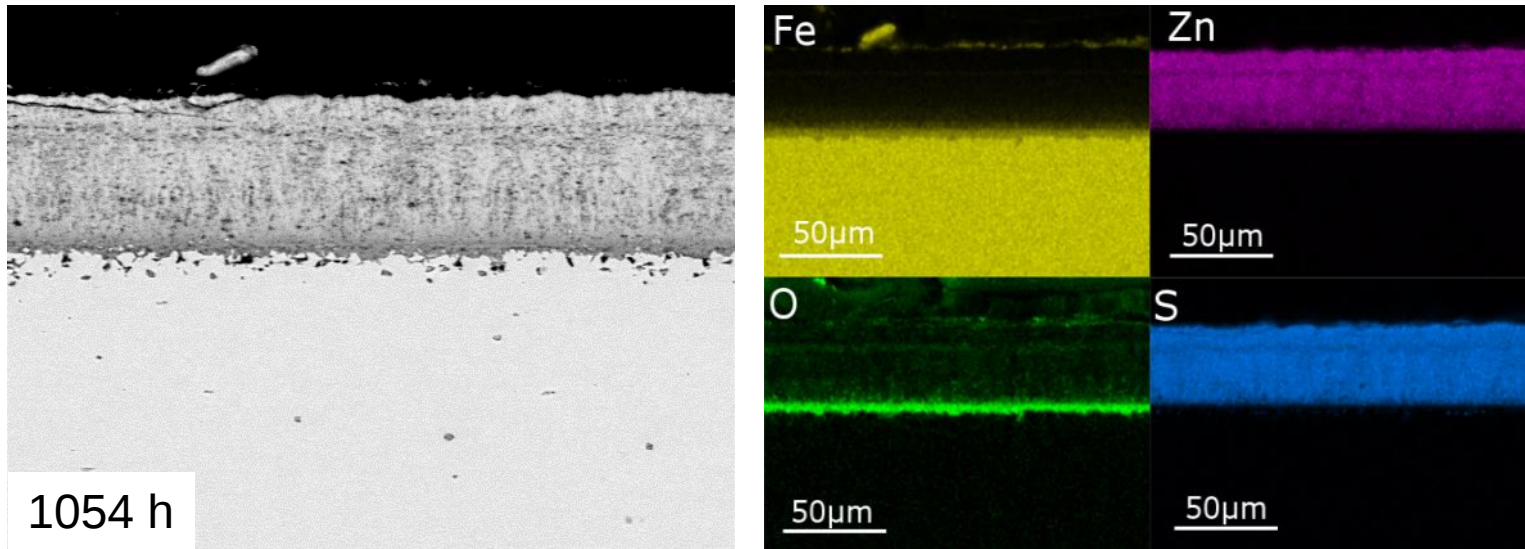
580 h



1084 h

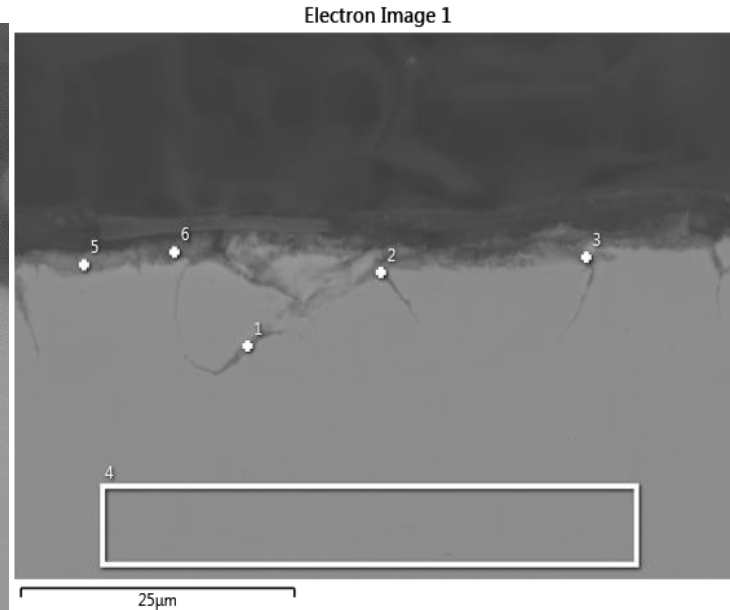
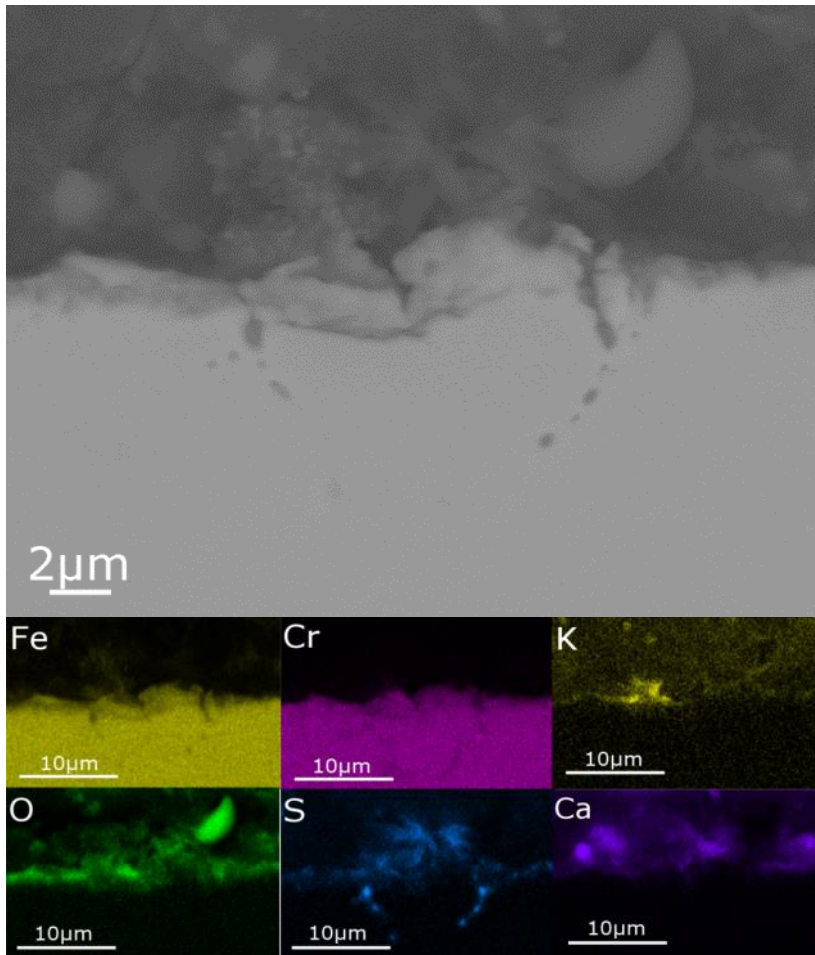
- All exposures show presence of CaCO_3 and ZnS
- K_2SO_4 signal only after intermediate exposure
- Ferrite phase clearly distinguishable for intermediate exposure

16Mo3 – long-term exposure



- Fe and O enriched at the metal/deposit interface.
- Zn and S abundant in outer deposit

304L – 1054 h



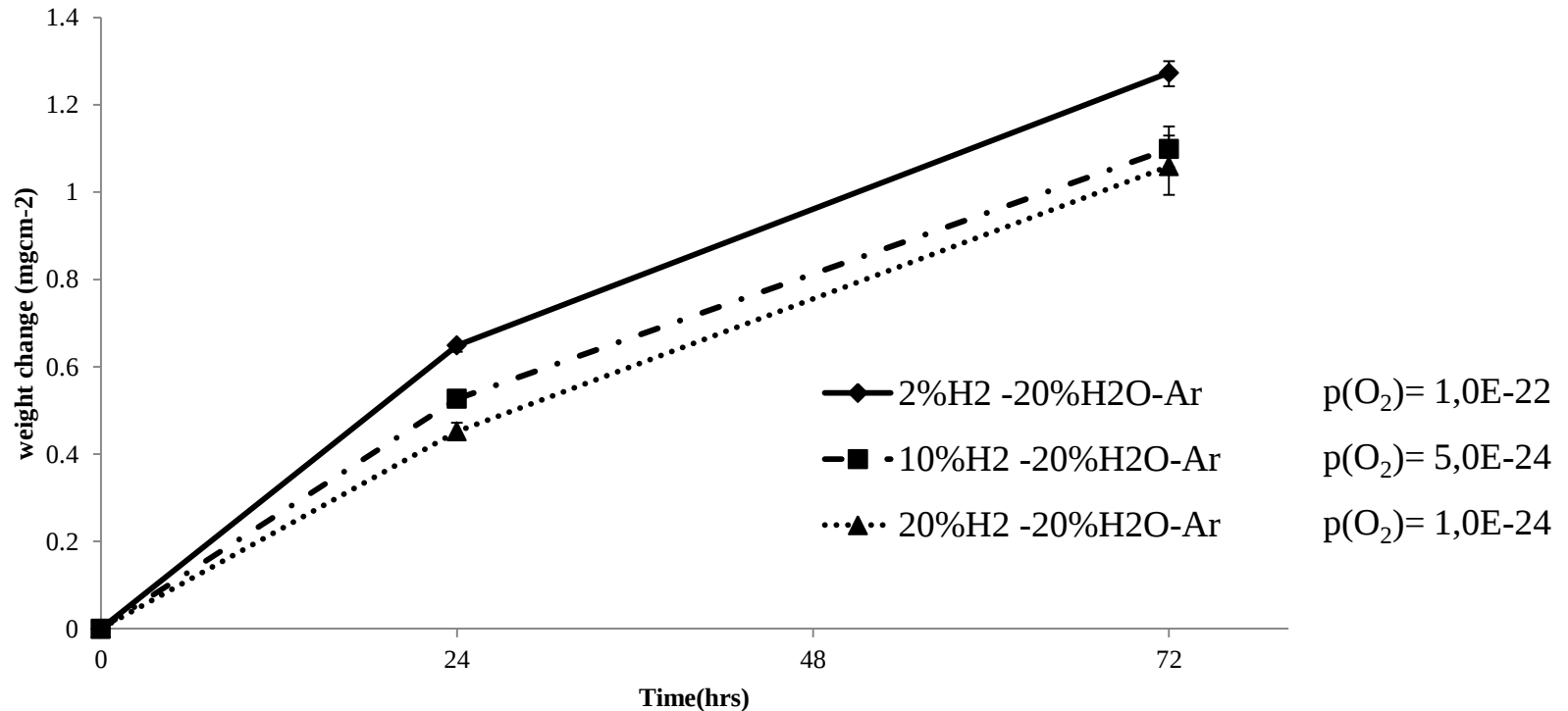
- Both inner oxidation and sulphidation
- Oxidation down to 10 μm , sulphidation to 7 μm
- Samples show deposit on surface

Field exposures – final remarks

- Exposures were performed below 500°C and with temperature variations
- Cl levels of the fuel were present in levels comparable to literature. No Cl was detected on the samples
- Metal loss small for 16Mo3
 - Low alloyed steel shows potential at this temperature
- All materials show signal from deposit of ZnS and CaCO₃ (XRD)
- Zn was present in significant amounts on the surfaces. Zn is proposed to affect the materials' behaviour
 - It is argued that zinc plays a key role in capturing sulphur in this environment, thus preventing iron from sulphidation.

Lab. Exposure

The Effect of H₂



- Samples : 304L
- Temperature= 600 °C

Composition [wt%]	Cr	Al	Mo	C	Mn	Si	Minor Add.	Fe	Ni
304L	18,18	-	0,53	0,022	1,53	0,31	Present	Bal.	8,10

Lab. Exposure The Effect of H₂

24 hours

72 hours

Increase in Time

2% H₂ – 20% H₂O

Epoxy (A)

Steel 20 μm

Epoxy (B)

Steel 20 μm

10% H₂ – 20% H₂O

Epoxy (C)

Steel 20 μm

Epoxy (D)

Steel 20 μm

20% H₂ – 20% H₂O

Epoxy (E)

Steel 20 μm

Epoxy (F)

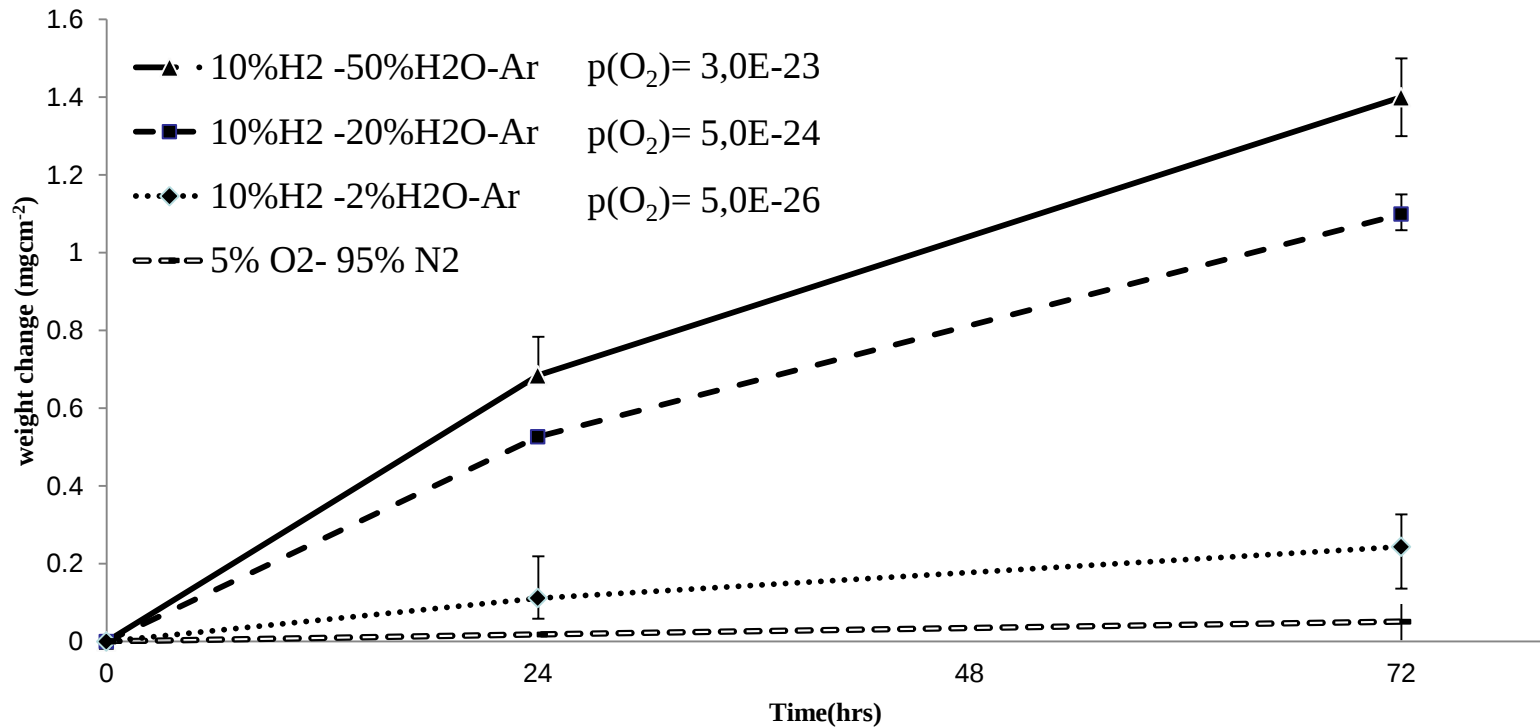
Steel 20 μm

Increase in oxide thickness

Increase in H₂ content

Lab. Exposure

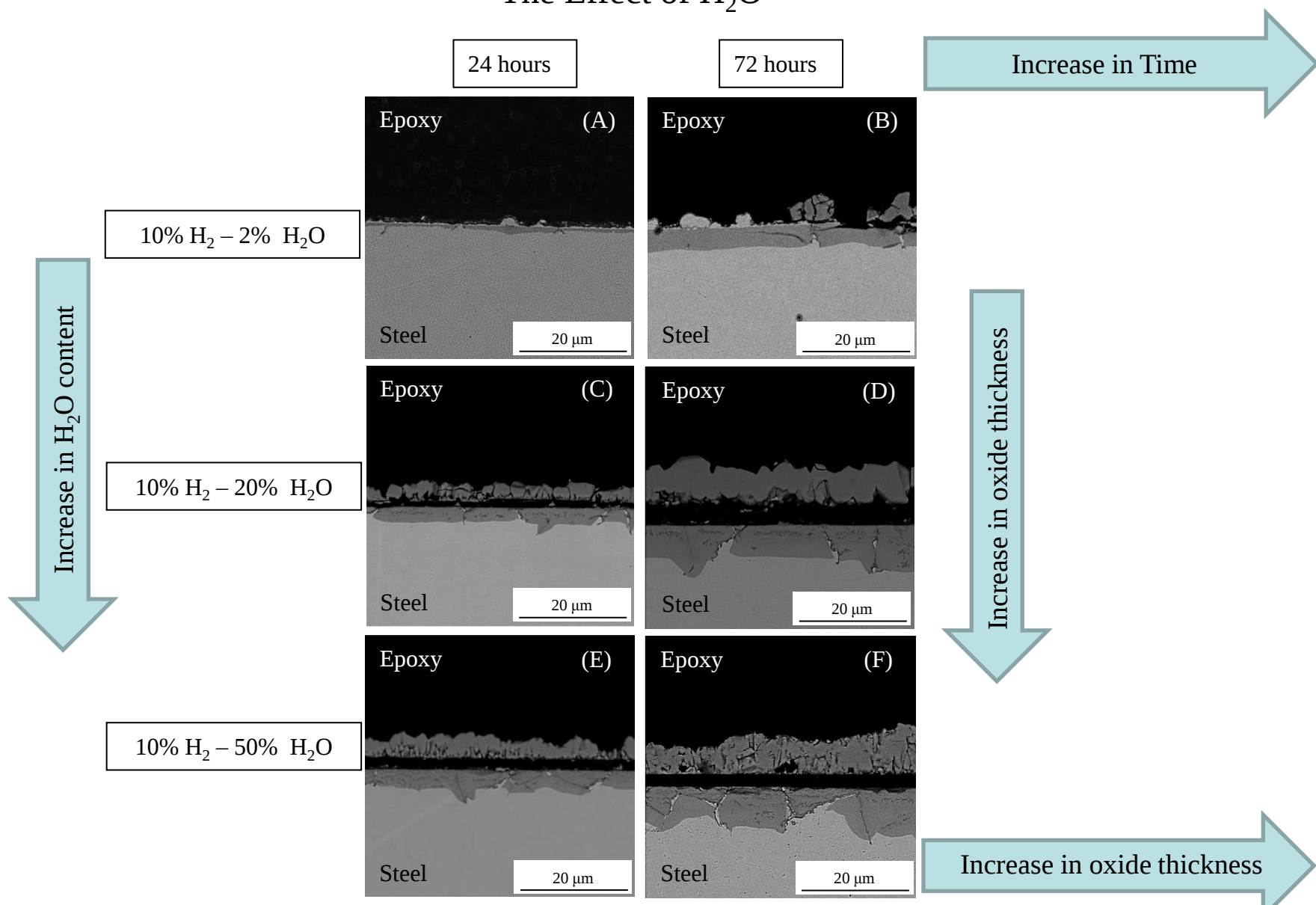
The Effect of H₂O



- Samples : 304L
- Temperature= 600 °C

Composition [wt%]	Cr	Al	Mo	C	Mn	Si	Minor Add.	Fe	Ni
304L	18,18	-	0,53	0,022	1,53	0,31	Present	Bal.	8,10

Lab. Exposure The Effect of H₂O



The Effect of H_2 and H_2O

Final remarks

- Marginal chromia formers, e.g. 304L, can be subject towards accelerated corrosion in low oxygen activity environments.
- Breakaway may be triggered by a deep chromium depletion of the alloy substrate.
- The chromium depletion was deeper in H_2 - H_2O environment than it is in, e.g. dry O_2 , because the pure chromia scale formed in H_2 - H_2O environment grew faster than the $Cr_{1-x}Fe_x)_2O_3$ scale formed in air and dry O_2 .

Ongoing work

- Material exposures have been made in the Cortus' gasifier in Köping
 - Very high temperature, $\sim 1100\text{ }^{\circ}\text{C}$
- Samples were recently removed and are about to be investigated



Thank you for your attention!

- We would like to acknowledge:

